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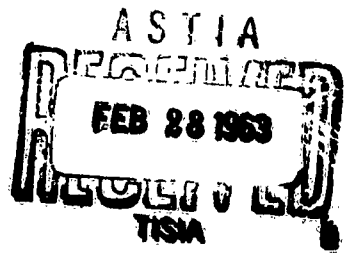
297022 *Refractomet Division*

UNIVERSAL-CYCLOPS STEEL CORPORATION

*Technical Report*

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Bridgeville, Pennsylvania

ASC TR 7-827 (VI)

ASC INTERIM REPORT 7-827 (VI)  
February, 1963

TUNGSTEN SHEET ROLLING PROGRAM

Contract AF33(600)-41917

Sixth Interim Report

1 September 1962 - 31 January 1963

Prepared By  
W. J. Schoenfeld

UNIVERSAL-CYCLOPS STEEL CORPORATION  
REFRACTOMET DIVISION  
BRIDGEVILLE, PENNSYLVANIA

The melting program was scaled up to provide 6<sup>in.</sup> conditioned ingots for extrusion. Direct extrusion to sheet bar and extrusion to round for subsequent press forging to sheet bar was accomplished satisfactorily on four ingots. The round extrusions were forged to sheet bar and rolling was initiated.

BASIC INDUSTRY BRANCH  
MANUFACTURING TECHNOLOGY LABORATORY

Directorate of Materials and Processes  
Aeronautical Systems Division  
United States Air Force  
Wright-Patterson Air Force Base, Ohio

ABSTRACT - Summary

Sixth Interim Technical Progress Report

ASC INTERIM REPORT 7-827 (VI)

February, 1963

# TUNGSTEN SHEET ROLLING PROGRAM

Refractomet Division  
Universal-Cyclops Steel Corporation

During the period of this report, melting and extrusion was scaled up to meet the sheet size requirements of this phase. Specifically, four 8" ingots were melted, conditioned to 6" extrusion billets. Two were extruded to 3" round for subsequent forging to sheet bar and the remaining two were extruded direct to 1.75" x 4" sheet bar. All extrusions were evaluated and the 3" rounds press forged to 2" thick sheet bar.

The first rolling operation was accomplished successfully on both press forged and direct extruded rounds.

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## FOREWORD

This Interim Technical Progress Report covers the work performed under Contract AF33(600)-41917 from 1 September 1962 to 31 January 1963. It is published for technical information only and does not necessarily represent the recommendations, conclusions, or approval of the Air Force.

This contract with the Refractomet Division, Universal-Cyclops Steel Corporation, Bridgeville, Pennsylvania, was initiated under ASC Aeronautical System Division, Project 7-827, "Tungsten Sheet Rolling Program." It was administered under the direction of Mr. Hugh L. Black, Project Engineer, Basic Industry Branch, Manufacturing Technology Laboratory, Wright-Patterson Air Force Base, Ohio.

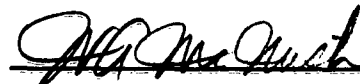
W. J. Schoenfeld of the Development Group, Refractomet Division, Universal-Cyclops Steel Corporation was the Engineer in charge.

Since the nature of this work is of interest to so many fields of endeavor, your comments are solicited as to the potential utilization of the material produced under this contract. In this manner, it is felt that a full realization of the resultant material produced will be accomplished.

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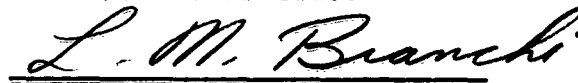
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## TABLE OF CONTENTS

	Page
I     Introduction	1
II    Phase IV - Program Objectives	1
III   Ingot Melting	2
IV    Extrusion	7
V     Sheet Bar Forging	15
VI    Sheet Rolling	24
VII   Conclusions	24

### List of Tables

I     Ingot Melting and Processing	3
II    Electrode Chemical Analyses	6
III   Ingot Chemical Analyses	6
IV    Extrusion Data for 6" Diameter Billets	8
V     Physical Dimensions of Extrusions	10
VI    Yield Summary from Extrusion Billet to Sheet Bar	22
VII   Reduction Schedule for Initial Rolling	24

### List of Figures

1     As-Cast 8" Diameter Ingot	4
2     As-Extruded 3" Diameter Rounds	9
3     As-Extruded 1-3/4" x 4" Cross Section Sheet Bar	11
4     Ultrasonic Evaluation of Extruded Rounds	13
5     Macrostructure of As-Extruded 3" Diameter Rounds	14
6     Macrostructure of As-Extruded Sheet Bar	16
7     DPH Hardness on KD1147 As-Extruded	17
8     DPH Hardness on KD1167 As-Extruded	18
9     DPH Hardness on KD1148 As-Extruded	19
10    DPH Hardness on KD1168 As-Extruded	20
11    Sheet Bar Forging Schedule	21
12    Press Forged Sheet Bar	23

## I. Introduction

The continued interest and support of government agencies in refractory metal systems has in the past few years resulted in a revolutionary advance in their technology. The utilization of this technology is rapidly expanding our arsenal of intercontinental weapons and eliminating man's barrier to space exploration.

This program, supported through the Aeronautical Systems Division, USAF, is designed to further the existing state-of-the-art in producing tungsten sheet. The information derived should provide a significant advance in the missile and space vehicle programs. To accomplish the overall mission, the program has been divided into five separate phases as summarized below.

Phase I	State-of-the-Art Survey—Report Issued
Phase II	Ingots Development—Report Issued
Phase III	Development of Rolling Operations—Report Issued
Phase IV	Process Uniformity Verification and Post-Rolling Development—In Process
Phase V	Final Pilot Production

## II. Phase IV - Program Objectives

The objectives of this phase are threefold:

### A. Verify Process Uniformity

Using the sheet rolling parameters established in Phase III, several sheets of each gauge will be rolled. Physical and mechanical properties will be determined and compared to establish the degree of control which can be expected.

#### B. Scale-up to 36" x 36" Sheet

All efforts in this phase are toward processing 36" x 36" sheet. Many problems usually evolve when wider widths are attempted. The extent to which these problems will occur on this program can only be determined through rolling experience.

#### C. Post-Rolling Development

Goals which have been established for the physical quality of the final material are shown below. Although all are affected by the actual rolling operation, additional post-rolling practices will be utilized in an attempt to meet these goals.

1. Surface finish—Number 2 Matte
2. Gauge Control— $\pm$  5%
3. Flatness—4% per MAB 176-M
4. Gauge Tolerance—1/2 of AMS 2242

In initiating work on this phase, a scale-up of the melting and extrusion practices were required and most of the effort during this period was expended in this area.

### III. Ingot Melting

The minimum ingot size requirement in this phase was conditioned 6" diameter in order to achieve an extrusion with a cross section compatible with the sheet size requirements. In Phase II an attempt was made to scale-up, however that resulted in

complete failure principally due to melting furnace deficiencies. An additional problem at that time was starting material technology. The quality requirements of electrodes were not known and methods of assembling (joining) electrodes in the furnace had not been established.

Extensive modifications were made to the arc melting furnace. These included additional power and modified power input, modified cooling and a change in the electrode feed mechanism. Concurrently, an investigation of electrodes was completed and a quality specification written. Machining investigations were run and satisfactory threading procedures established. Connecting nipples were made by extruding small ingots to 1-5/8" diameter and subsequently threading these.

Four ingots were melted into an 8" ID mold. The melting conditions were essentially satisfactory, however, at intervals the melt became very erratic. This was attributed to the basic electrode as the conditions would initiate when proceeding from one bar to the next, and would stop when this bar was consumed and the melting of the next bar initiated. A typical as-cast ingot is shown in Figure 1. The melting history and billet yields for these ingots is shown in Table I.

TABLE I  
Ingot Melting and Processing  
Heat Number

	<u>1147</u>	<u>1148</u>	<u>1167</u>	<u>1168</u>
Mold Dia. (in.)	8	8	8	8
Electrode Dia. (in.)	3-1/4	3-1/2	3-7/8	3-7/8
Weight Melted (#)	518	523	571	453
Cond. Weight (#)	231	233	301	153
Yield %	44.7	44.6	52.8	33.8



FIGURE 1 - AS-CAST 8" DIAMETER INGOT

It will be noted from the table that the yields were relatively low. This is due largely to the fact that, in order to insure a completely satisfactory ingot at the required 6" diameter, an 8" mold was used. In observing the machining of these ingots, sidewall porosity was eliminated in every case at 7" to 7-1/4", so that a much higher yield could have been realized if a 7" extrusion container were available. The other area of appreciable yield loss was on the hot top of three of the four ingots. On these three the average yield loss for hot top cropping only was 14.5%.

Chemical analysis of the starting electrode is listed in Table II. Table III lists the ingot chemistries. In comparing the two tables, two elements deserve discussion. In the second electrode powder lot the nickel content is relatively high and well above the 20 PPM maximum specification level. Rather than reject the material because of considerable time delay, the material was melted subject to rejection if the ingot chemistry were not satisfactory. As shown in the ingot chemistry for Heats KD1167 and 1168, the nickel content using these electrodes was below the 1 PPM detection limit. The molybdenum content in the two powder lots is shown to be 8 and 11 PPM respectively. In the ingot chemistries only one heat is below 100 PPM. This large deviation between electrode and ingot chemistry has been a continuing problem, yet unresolved. Heats 1147 and 1148 were both melted using the first powder lot, yet the molybdenum content is 100 and 500 PPM respectively. The remaining two heats were melted using the second powder lot and the ingot analyses are <10 and 190 PPM respectively.

It would appear from the work to date that the molybdenum in the powder lot is not uniform, however, analyses of the powder lot are consistently low.

TABLE II  
Electrode Chemical Analyses

Lot	Element											
	As	Al	Ca	Cr	Cu	Fe	K	Mg	Mo	Na	Ni	Si
964	<3	0.6	6	2	0.1	4	<20	<2	8	7	6	3
996	<3	9	6	18	0.1	14	35	<2	11	20	45	3

TABLE III  
Ingot Chemical Analyses

Lot	Element															
	Mn	Al	V	Cr	Cu	Fe	Co	Mg	Mo	Ti	Ni	Si	C	O	N	H
1147 <sup>1</sup>	<10	<10	<10	<10	<1	13	<5	<1	100	<1	<1	<20	30	14	20	1.9
1148 <sup>1</sup>	<10	<10	<10	<10	<1	13	<5	<1	500	<1	<1	<20	30	11	21	1.6
1167 <sup>2</sup>	<10	<10	<10	<10	<1	12	<5	<1	<10	<1	<1	<20	10	9	2	<1
1168 <sup>2</sup>	<10	<10	<10	<10	<1	5	<5	<1	190	<1	<1	<20	44		14	

- 1 - Electrodes from powder lot 964  
2 - Electrodes from powder lot 996

All analyses in PPM

(<) Indicates analysis was below detection limits

Based on the fact that these initial ingots were free of porosity at 7" to 7-1/4", it was concluded that a nominal as-cast 8-5/8" ingot should "clean-up" at approximately 7-7/8" which would be satisfactory for the 8" extrusion container. The existing 8" ID mold was machined to 8-5/8" ID and two additional ingots melted. No data are available on these two ingots as they have not been machined. It is doubtful if they will be free of porosity at 7-7/8", however, as the cooling shrinkage was much greater than anticipated. Both ingots as-cast were 8-1/8" to 8-1/4" diameter so that the shrinkage was a nominal 1/2" on the diameter. This compares with a nominal 1/4" shrinkage on the previous 8" ingots.

#### IV. Extrusion

Four billets were extruded on the du Pont 2750 ton press. Two were extruded to 3" diameter rounds for subsequent press forging to sheet bar and the other two were extruded directly to 1-3/4" x 4" cross section sheet bar. Table IV lists the extrusion parameters used and the resultant pressure requirements. As shown, all billets were extruded at a Shawmeter temperature of 3200°F. based on the successful results of 4" billets extruded at this temperature. The break-through pressure requirements were relatively consistent except for the first extrusion which was somewhat higher. It is shown by the running pressure that the sheet bar extrusions require slightly more pressure than the rounds. The average extrusion constant of the previous 4" billets was 81,600 psi. As shown, this is lower than the first billet but higher than the other three.

Figure 2 shows the two as-extruded rounds after sandblasting, with the end cropping requirements, determined by contact ultrasonic, indicated. The relatively large amount to be cropped off



TABLE IV  
Extrusion Data for 6" Diameter Billets

Heat Number	Billet Weight	Temperature °F. Shaw Optical	Pressure - psi Breakthrough Running	Speed IPS	Extrusion Constant (K)
1147	231	3200	118,000	16.5	85,000
1148	233	3200	108,000	16	78,000
1167	303	3200	100,000	17	72,250
1168	153	3200	108,000	18	78,000

Extrusion Constant  $K = \frac{P}{A \ln \frac{A}{a}}$  where:  
P = maximum pressure in pounds  
K = extrusion constant in psi  
A = cross sectional area of container  
a = cross sectional area of extrusion



FIGURE 2 - AS-EXTRUDED 3" DIAMETER ROUNDS

the tail (nominal 4") is due to a deep tail pipe but actually does not represent a completely solid piece since this area is hollow. The picture shows that the general surface was excellent and it should also be noted that no die wash occurred. Actually, both billets were pushed through the same die which had not been accomplished previously. An as-extruded sheet bar is shown in Figure 3. This picture shows also that no die wash occurred which is remarkable considering the sharp corner angles required on the sheet bar die. The fact that no die wash occurred on these extrusions is further verified by the physical dimensions shown in Table V.

TABLE V  
Physical Dimensions of Extrusions

Cross Section	Heat Number			
	1147	1148	1167	1168
Nose	3.034"Dia.	3.984"x 1.767"	3.043"Dia.	3.975"x 1.765"
Tail	3.018"Dia.	3.990"x 1.765"	3.030"Dia.	3.975"x 1.768"
As-Extruded Length	48	49	59	32

To more accurately evaluate the internal quality of the round extrusions, immersion ultrasonic evaluation was required. In order to accomplish this, they had to be straightened. They were, therefore, heated to 2300°F. in a hydrogen atmosphere furnace, straightened on a 1500 ton press and subsequently reheated to 2300°F., soaked for ten minutes and then buried in vermiculite.

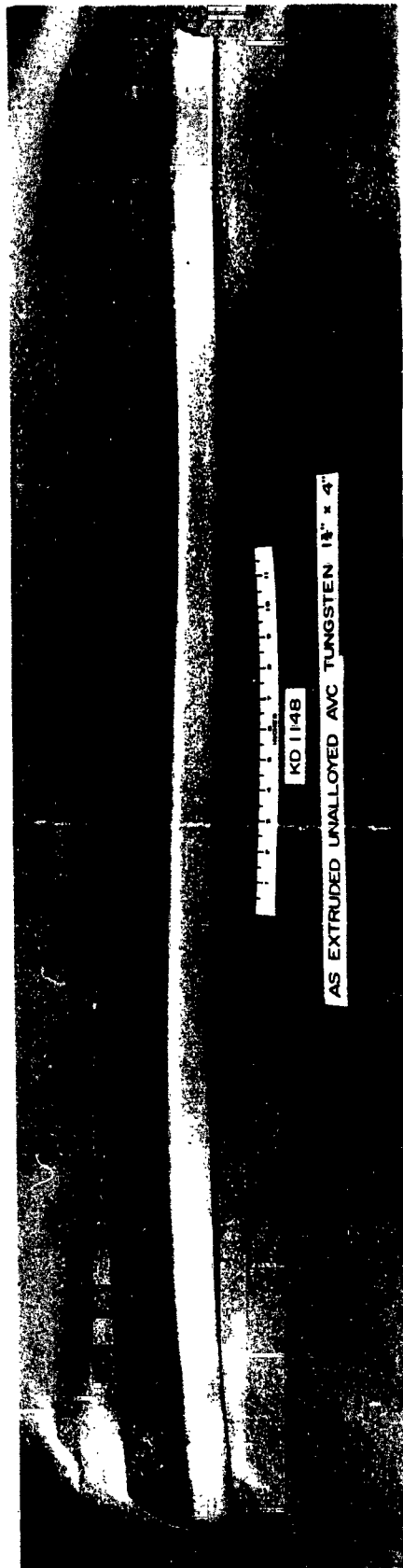
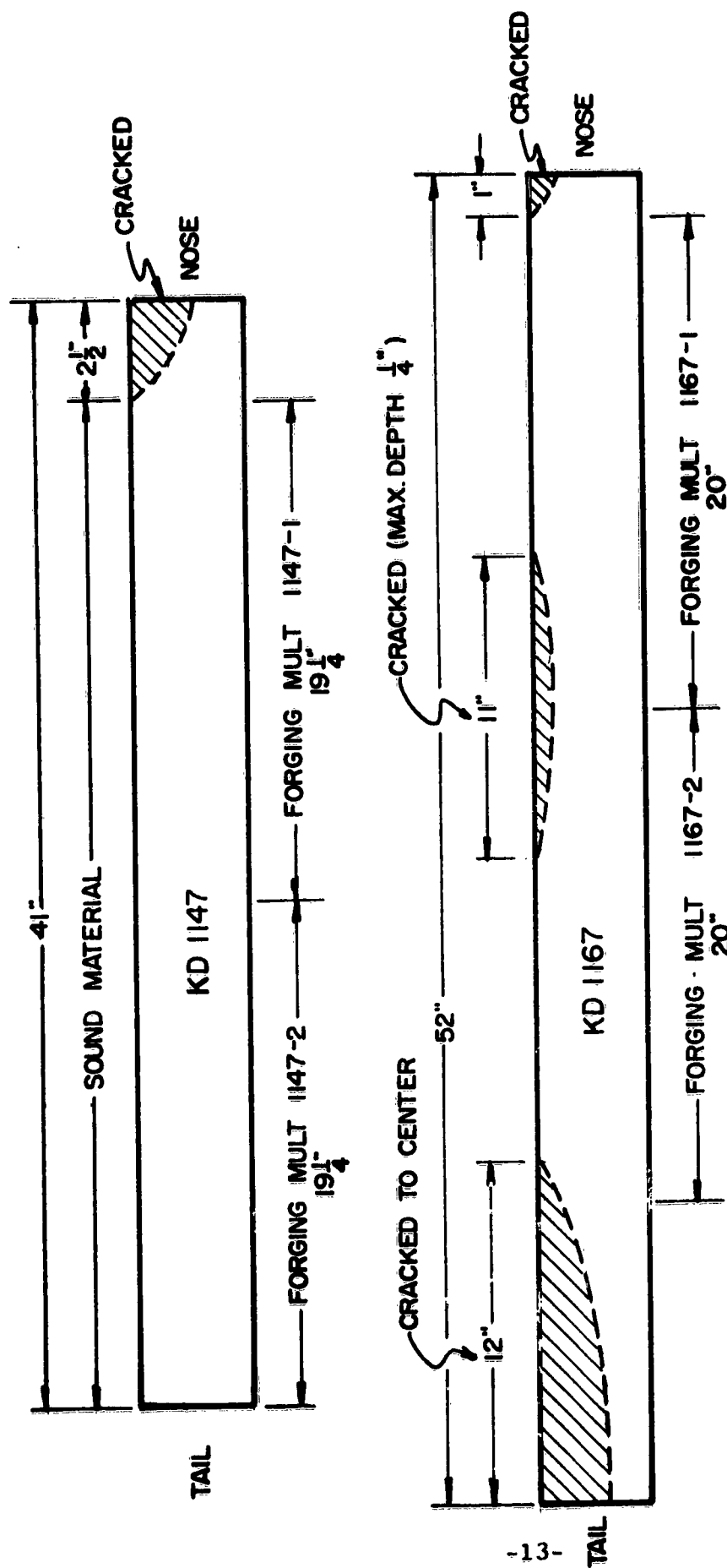


FIGURE 3 - AS-EXTRUDED  $1\text{-}3/4\text{'}$  x  $4\text{'}$  CROSS SECTION SHEET BAR

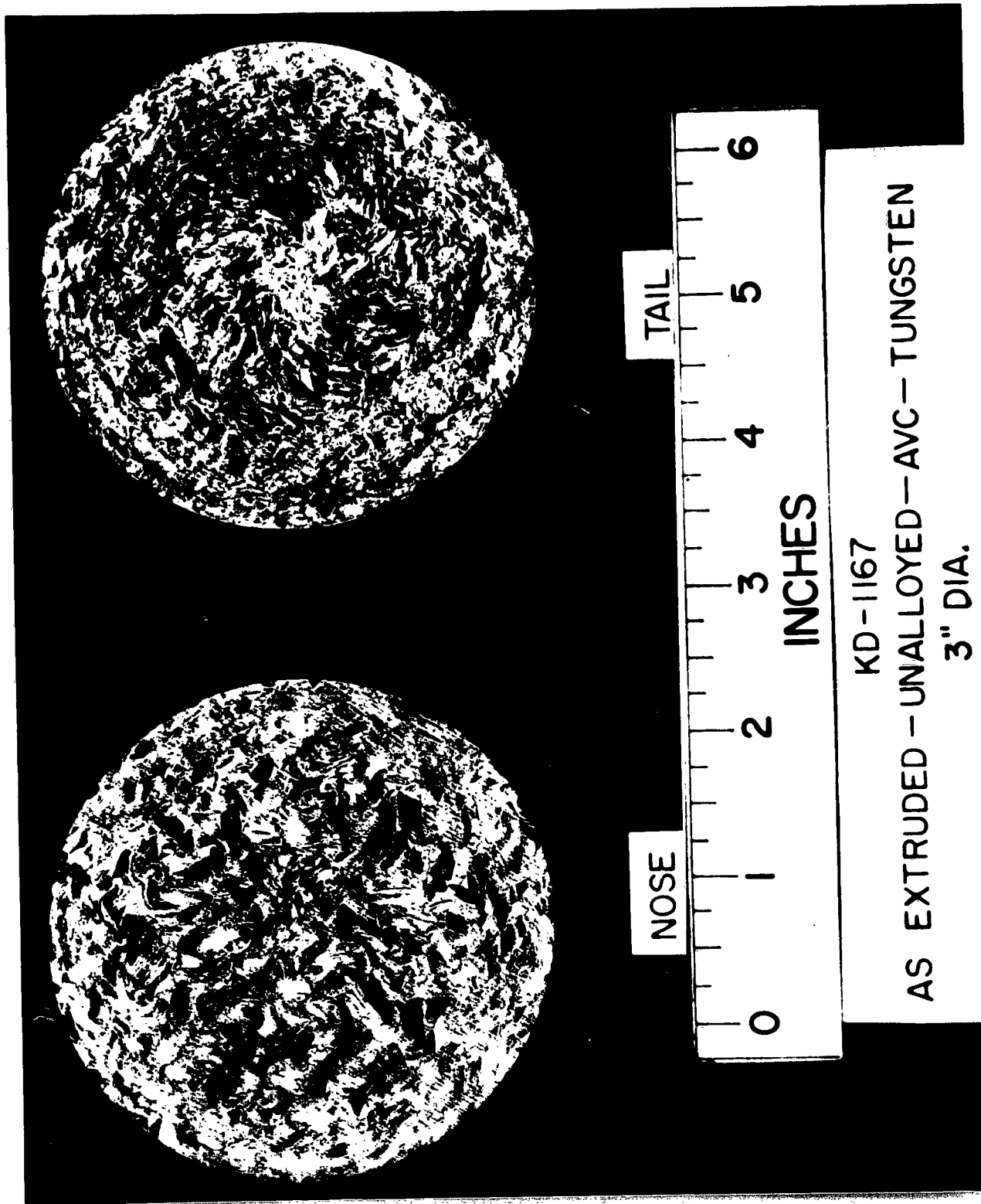
Immersion ultrasonic examination indicated that on both extrusions a longitudinal crack varying in depth up to  $1/2$ " extended along the entire length, however, the results were not precise due to slight surface defects. In addition, 12" on the trailing end of 1167 appeared to be cracked from surface to center. As the surface defects were preventing an accurate evaluation, the extrusions were machined to 2.850" and then surface ground to 2.830". They were immersion ultrasonic examined again and the results are plotted in Figure 4. As shown in this figure, 1147 was free of defects except for 2.5" on the nose end. On 1167, 12" on the trailing was cracked tapering from the surface to the center. In addition, 11" was cracked to a maximum depth of  $1/4$ " and 1" on the nose was cracked to a depth of  $1/2$ ". As shown on this diagram, two forging mults were cropped from each extrusion. The cracked areas on 1167 were ground out prior to forging.

After cropping the as-extruded sheet bar, minor surface conditioning was required to prepare it for subsequent rolling. This extrusion was also cropped into two mults for the rolling operation.

Macro discs were cropped from the ends of all extrusions. The macro structure of 1167 is shown in Figure 5. Note that the nose section has a much larger grain size than the tail. This can be attributed to the fact that the extreme nose end of a billet in proceeding through the extrusion die is not worked, particularly in the center, to the extent of material back several inches from the nose.



**FIGURE 4**  
ULTRASONIC EVALUATION OF EXTRUDED ROUNDS



KD-1167  
AS EXTRUDED - UNALLOYED - AVC - TUNGSTEN  
3" DIA.

FIGURE 5 - MACROSTRUCTURE OF AS-EXTRUDED 3" DIAMETER ROUND

Figure 6 shows the macrostructure of the as-extruded sheet bar. The nose section shows a relatively equiaxed grain structure and the tail shows a wrought fibrous structure. The equiaxed structure in the nose is probably related to its close proximity to the area of initial deformation thus producing the same effect discussed for the extruded rounds.

A hardness survey was made on the nose and tail of each extrusion. This information is plotted in Figures 7, 8, 9 and 10. In comparing the two extruded rounds, it will be noted that the average hardness of the tails is within 1 DPH. The nose of 1147 is slightly harder than 1167 which can be attributed to the fact that 1167 hardness values were taken closer to the nose and therefore represent material with less work. As shown, the sheet bars are harder than the rounds. This can be attributed to a higher degree of work related to the sheet bar configuration.

#### V. Sheet Bar Forging

For the forging operation, a 1500 ton hydraulic press was used in conjunction with a hydrogen atmosphere furnace. As this press was relatively slow acting, the pieces were heated to 2600°F. in order to maintain a nominal 2000°F. forging temperature. The actual forging process used was as follows:

1. Charge forging mult into 2600°F. furnace;
2. Soak five (5) minutes after reaching temperature;
3. Transfer to press and forge 3/4" flats as shown in Figure 11;
4. Reheat to temperature and hold five minutes;
5. Transfer to press, rotate 90° to initial forging direction and forge to nominal 2" thick;
6. Reheat to temperature, hold 10 minutes, discharge and bury in vermiculite.



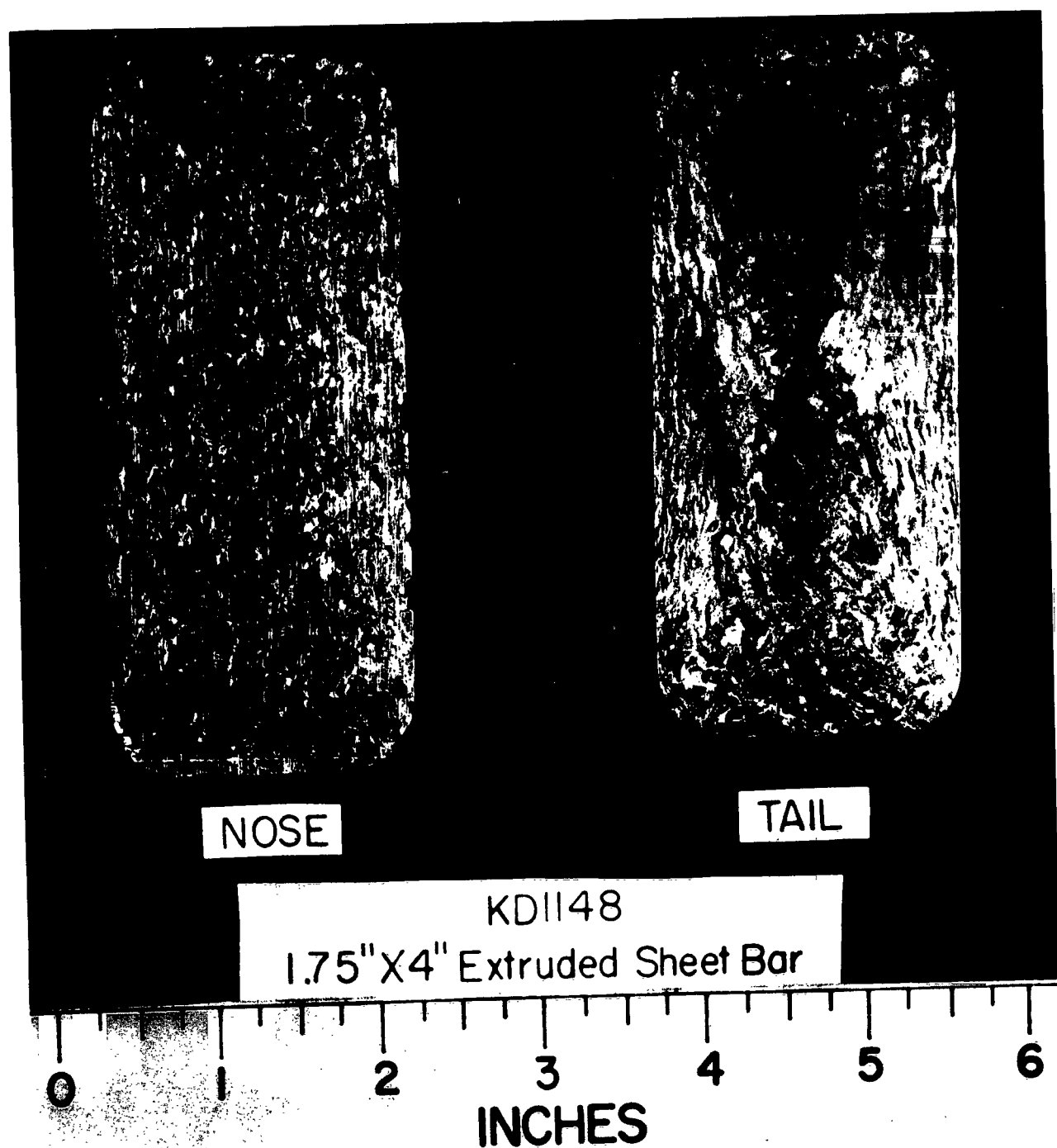
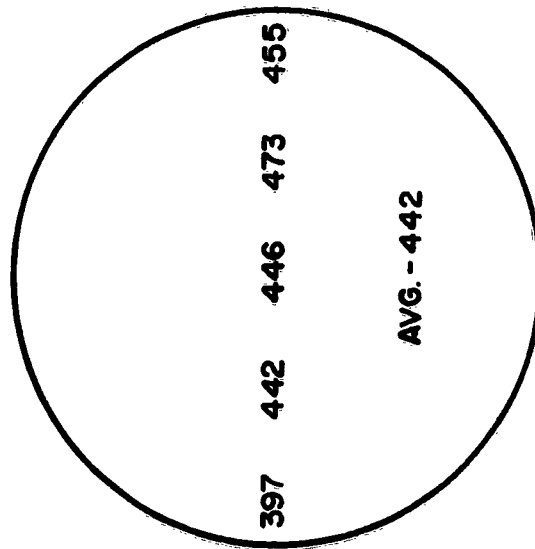
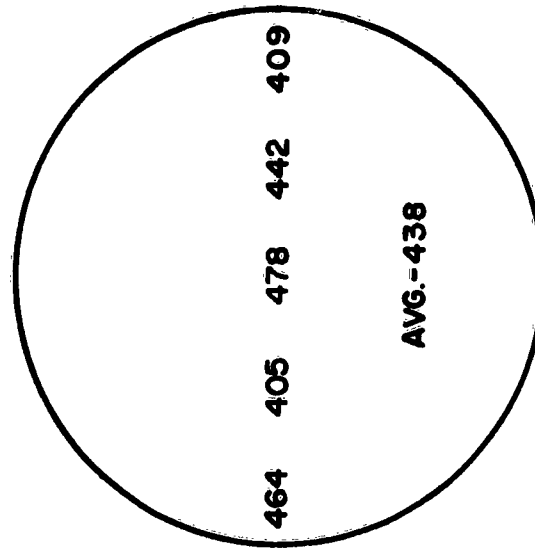


FIGURE 6 - MACROSTRUCTURE OF AS-EXTRUDED SHEET BAR

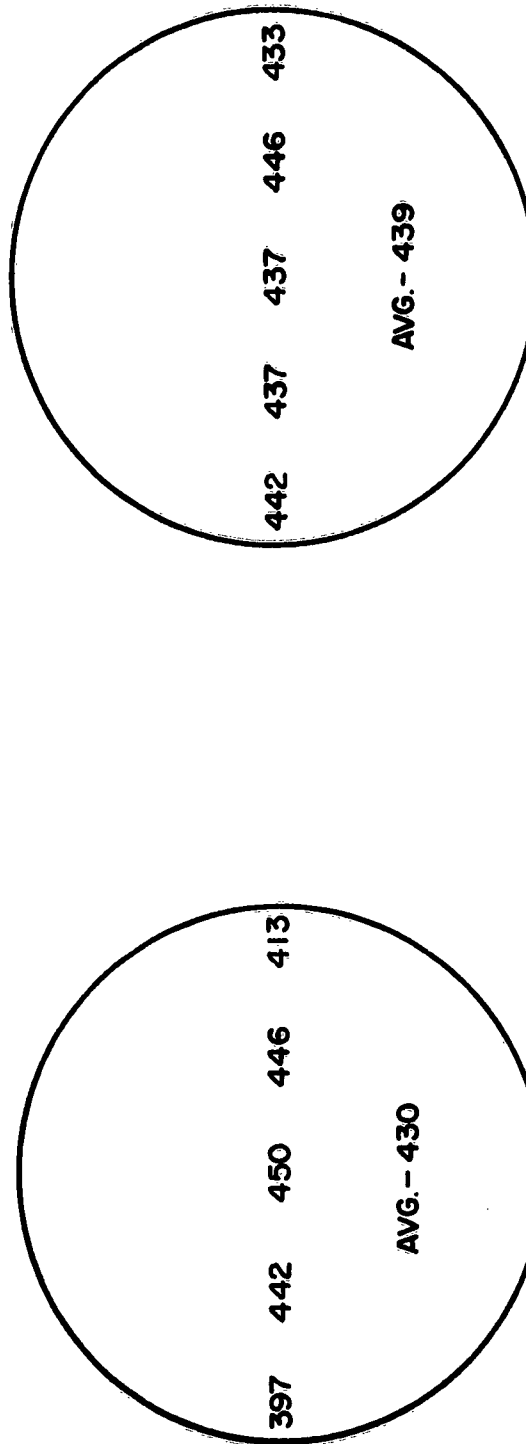


NOSE



TAIL

**FIGURE 7**  
**DPH HARDNESS ON KD 1147 AS-EXTRUDED**  
**(10 KG LOAD)**



**FIGURE 8**  
**DPH HARDNESS ON KD 1167 AS-EXTRUDED**  
**(10 KG LOAD)**

433	446	446	464
405	421	437	464
AVG.-439			

NOSE

455	429	380	387
446	473	464	405
AVG.-430			

TAIL

**FIGURE 9**  
 DPH HARDNESS ON KD 1148 AS-EXTRUDED  
 (10 KG LOAD)

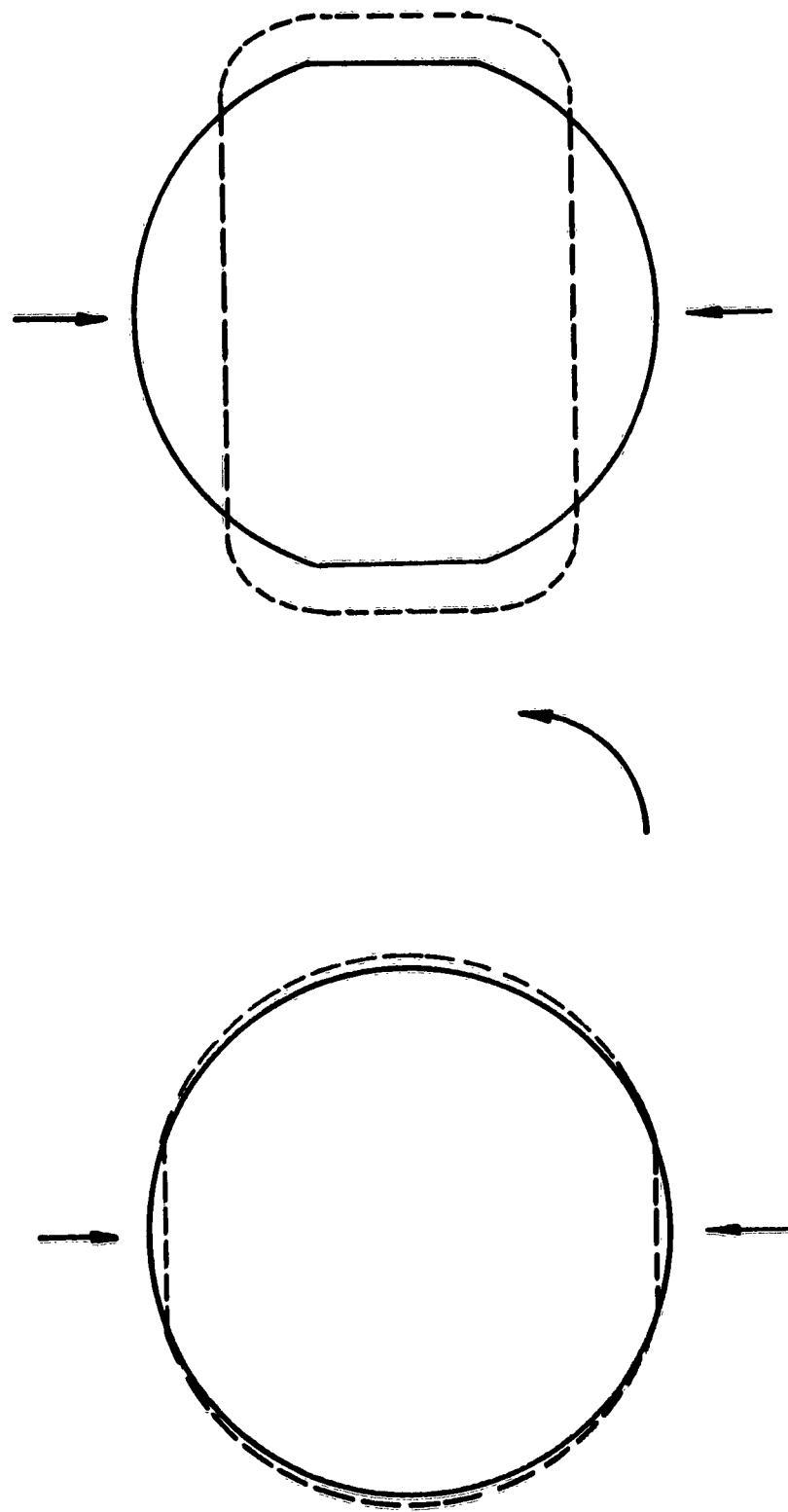
465	435	428	462
475	441	448	468
AVG. - 453			

NOSE

480	475	431	475
485	442	468	472
AVG. - 466			

TAIL

**FIGURE 10**  
**DPH HARDNESS ON KD 1168 AS-EXTRUDED**  
**(10 KG LOAD)**



PRESS 1" FLATS — REHEAT — ROTATE 90° — FORGE TO 2" THICK

- REPRESENTS INITIAL CROSS SECTION
- - - REPRESENTS CROSS SECTION AFTER FORGING OPERATION

**FIGURE 11**  
**SHEET BAR FORGING SEQUENCE**

In Step 5 above, the press stalled out at a nominal 2-3/16" thick. The first mult was reheated to temperature and an attempt made to forge it down to 2", however, only 1/16" additional reduction was achieved. The remaining three pieces were only forged once in Step 5. After slow cooling, the pieces were sand blasted for inspection. The four mults are shown in Figure 12. Although no cracks are visible, closer inspection showed light surface ruptures on all of the pieces. Two of the pieces had one larger crack running parallel to the extrusion direction. These were probably related to the cracks initially indicated on the extrusions but which were supposedly removed by machining and grinding. It is suggested that these cracks were present in the conditioned extrusion but the depth was so minor that ultrasonic inspection did not pick them up. Although the pieces had also been dye penetrant inspected, flowed metal on the surface prevented detection by this method. These cracks in both mults were conditioned out at a depth of 1/8". The remaining surface of these two and the remaining two were ground lightly to remove the light surface ruptures. The yield losses on extrusion and forging are presented in Table VI.

TABLE VI  
Yield Summary from Extrusion Billet to Sheet Bar

Heat No.	Extruded Weight	End Loss	Surface Loss	Forging Mult Weight	Cond. Sheet Bar	% Yield Extrusion to Sheet Bar
1147	231	23-1/4	33-1/2	168 2 pcs.	145 2 pcs.	62.8
1148*	233	25	12	--	192 2 pcs.	82.5
1167	301	72	35	180-1/2 2 pcs.	160-1/2 2 pcs.	53.3
1168*	153	27-3/4	Not Completed			

\* Extruded directly to sheet bar

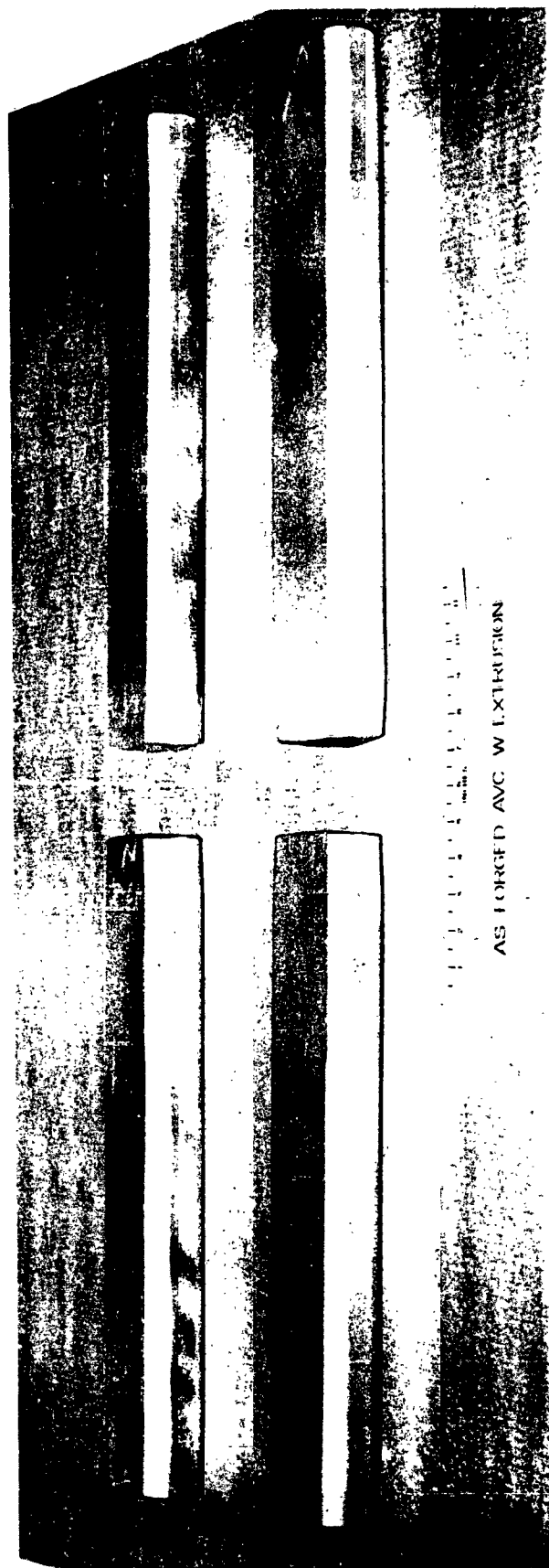


FIGURE 12 - PRESS FORGED SHEET BAR FROM EXTRUDED 3" DIAMETER ROUNDS



It will be observed immediately from the table that the yield in extruding directly to sheet bar is significantly higher and, in addition, eliminates the forging step. The low yield in 1167 is due in part to the cracked portions of the extrusion which in the table are included as end losses.

#### VI. Sheet Rolling

Two extruded sheet bar, mults 1148-1 and 1148-2, and two press forged sheet bar, 1167-1 and 1167-2, were rolled to an intermediate gauge of 1" using a 2300°F. furnace temperature. One pass per reheat was used for this initial rolling step with the reductions per pass shown in Table VII.

TABLE VII  
Reduction Schedule for Initial Rolling

Mult Code	Gauge After Pass Indicated					
	Initial	First	Second	Third	Fourth	Fifth
1148-1	1.70	1.50	1.32	1.15	1.025	---
1148-2	1.70	1.47	1.30	1.15	1.015	---
1167-1	2.00	1.75	1.52	1.32	1.15	1.015
1167-2	2.00	1.75	1.52	1.32	1.15	1.010

No visible cracking occurred during rolling. No further work was accomplished to date.

#### VII. Conclusions

- A. Arc cast tungsten ingots can be melted which will condition to 6" diameter extrusion billets.

- B. The extrusion of 6" billets to 3" diameter and 1.75" x 4" sheet bar can be accomplished satisfactorily; however, minor cracking problems did occur in the rounds which will require some modification to the extrusion practice.
- C. Sheet bar can be press forged from 3" diameter rounds in the temperature range of 2000°F.
- D. Initial rolling of both press forged and extruded sheet bar can be accomplished in the temperature range of 2300°F.

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Buffalo 5, New York

Bendix Products Division  
Bendix Aviation Corporation  
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401 N. Bendix Drive  
South Bend, Indiana

Boeing Airplane Company  
Attn: Mr. Edward Czarnecki  
Materials Mechanics and Structures  
Branch  
Systems Management Office  
P. O. Box 3707  
Seattle 24, Washington

Boeing Airplane Company  
Wichita Division  
Attn: Mr. W. W. Rutledge  
Mfg. Manager  
Wichita, Kansas

Ballistic Missiles Center  
Attn: Major A. F. Lett, Jr.  
P. O. Box 262  
Los Angeles 45, California

Bureau of Mines  
Albany, Oregon  
Attn: Mr. R. Beall

Bureau of Naval Weapons  
Department of the Navy  
Materials Branch (AER-AB-4)  
Attn: Mr. N. E. Promisel  
Washington 25, D. C.

Page 2 - Distribution List - Contract AF33(600)-41917

Chance Vought Aircraft, Inc.  
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Chief, Research and Development  
Dallas, Texas

Climax Molybdenum Company of Michigan  
14410 Woodrow Wilson Boulevard  
Detroit 3, Michigan  
Attn: Mr. George A. Timmons  
Director of Research

Commanding Officer  
Watertown Arsenal  
Attn: Mr. S. V. Arnold  
Watertown 72, Massachusetts

Convair-Division  
General Dynamics Corporation  
Attn: Mr. A. T. Seeman, Chief of  
Manufacturing Engineering  
P. O. Box 1011  
Pomona, California

Convair-Division  
General Dynamics Corporation  
Attn: Mr. J. H. Pamme, Director  
Manufacturing Development  
Mail Zone 2-22  
San Diego 12, California

Convair-Division  
General Dynamics Corporation  
Attn: Mr. W. O. Sunafrank  
Project Engineer  
Department 23-2  
Fort Worth, Texas

Curtiss-Wright Corporation  
Attn: Mr. O. Podell  
Vice President-Operational  
Planning  
304 Valley Boulevard  
Wood-Ridge, New Jersey

Curtiss-Wright Corporation  
Metals Processing Division  
Attn: Mr. V. T. Gorguze, Gen. Mgr.  
760 Northland Avenue  
Buffalo 15, New York

Douglas Aircraft Company, Inc.  
Attn: Production Design Engineer  
2000 N. Memorial Drive  
Tulsa, Oklahoma

Douglas Aircraft Company, Inc.  
Attn: Materials Division Group  
El Segundo, California

The Dow Chemical Company  
Attn: Mr. T. E. Leontis,  
Assistant to the Director  
Midland, Michigan

Firth Sterling, Incorporated  
3113 Forbes Street  
Pittsburgh 30, Pennsylvania  
Attn: Dr. C. H. Toensing

General Electric Company  
Attn: Mr. Louis P. Jahnke  
Manager, Metallurgical Engineering  
Applied Research Operations -  
Propulsion Laboratory  
Aircraft Gas Turbine Department  
Evendale, Ohio

Grumman Aircraft Engineering Corp.  
Manufacturing Engineering  
Attn: Mr. William J. Hoffman  
Vice President  
Bethpage, Long Island, New York

Aerojet General Corporation  
Attn: Mr. Alan V. Levy, Head  
Materials Research and Development  
Solid Rocket Plant  
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Sacramento, California

Ladish Company  
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5400 Packard Avenue  
Cudahy, Wisconsin

Page 3 - Distribution List - Contract AF33(600)-41917

Lockheed Aircraft Corporation  
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Lockheed Aircraft Corporation  
Attn: Mr. H. Fletcher Brown  
Manufacturing Manager  
Marietta, Georgia

Lockheed Aircraft Corporation  
Van Nuys, California

Lockheed Aircraft Corporation  
Missile Systems Division  
Attn: Mr. Clayton O. Matthews  
Sunnyvale, California

Lycoming Division  
AVCO Manufacturing Corporation  
Attn: Mr. W. A. Panke, Superintendent  
Manufacturing Engineer  
Stratford, Connecticut

Marquardt Aircraft Company  
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Director of Manufacturing  
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Van Nuys, California

Marquardt Aircraft Company  
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Manufacturing Engineer  
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Ogden, Utah

The Martin Company  
Attn: Chief Librarian  
Engineering Library  
Baltimore 3, Maryland

The Martin Company  
Denver Division  
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Materials Engineering  
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Denver 1, Colorado

Materials Advisory Board  
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2101 Constitution Avenue  
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McDonnell Aircraft Corporation  
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St. Louis 3, Missouri

National Aeronautics and Space  
Administration  
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Chief, Materials and Structures  
Division, Lewis Research Center

Navy Department  
Industrial Planning Division  
Attn: E. G. Gleason  
Washington 25, D. C.

North American Aviation, Inc.  
Attn: Mr. D. H. Mason  
Staff Engineering  
General Data Section  
International Airport  
Los Angeles 45, California

North American Aviation, Inc.  
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Materials Engineer  
International Airport  
Los Angeles 45, California

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Hawthorne, California

Page 4 - Distribution List - Contract AF33(600)-41917

Nuclear Metals, Inc.  
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Pratt & Whitney Aircraft Corporation  
CANEL, Connecticut Operations  
Attn: Mr. L. M. Raring, Chief  
Metallurgical and Chemical Laboratory  
P. O. Box 611  
Middletown, Connecticut

Reactive Metals, Inc.  
Attn: Mr. L. G. McCoy  
Government Contract Administrator  
Niles, Ohio

Republic Aviation Corporation  
Attn: Mr. Adolph Kastelowitz,  
Director of Manufacturing Research  
Farmingdale, Long Island, New York

Rocketdyne Division  
North American Aviation Corporation  
Department 574  
Attn: Mr. J. D. Hall  
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Rohr Aircraft Corporation  
Attn: Mr. Burt F. Raynes, Vice President  
Manufacturing  
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Chula Vista, California

Ryan Aeronautical Company  
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Vice President, Manufacturing  
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Sandia Corporation  
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Sikorsky Aircraft Division  
United Aircraft Corporation  
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Manager  
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Solar Aircraft Company  
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2200 Pacific Highway  
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Sperry Gyroscope Company  
Division of Sperry Rand Corporation  
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Engineering Librarian  
Great Neck, Long Island, New York

Sylvania Electric Products Corporation  
Attn: Dr. Paul Felton  
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Sylvania Electric Products Corporation  
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Bayside, New York

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Page 5 - Distribution List - Contract AF33(600)-41917

Thiokol Chemical Corporation  
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Materials and Processes Section  
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Titanium Metals Corporation of America  
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University of California  
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Wah Chang Corporation  
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Albany, Oregon

Westinghouse Electric Corporation  
Lamp Division  
Bloomfield, New Jersey  
Attn: Dr. R. H. Atkinson

Westinghouse Laboratories  
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Attn: Dr. J. H. Bechtold,  
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Wright-Patterson Air Force Base, Ohio

Wright Air Development Division  
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Mr. E. E. Zink  
Wright-Patterson Air Force Base, Ohio

Wright Air Development Division  
Attn: ASD (ASRCMP-4)  
Wright-Patterson Air Force Base, Ohio

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